

Improved impact strength of poly(lactic acid) by incorporating poly(butylene succinate) and silicon dioxide nanoparticles

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Abstract—The surface of silicon dioxide (SiO₂) nanoparticles was treated with oleic acid, and the resulting surface properties were characterized. Bio-based poly(lactic acid) (PLA)/poly(butylene succinate)/SiO₂ nanocomposites were fabricated via solution blending. The influence of the SiO₂ content on the thermal stability, flexural properties, impact strength, and morphology of the prepared nanocomposites was investigated using several techniques. The impact strength of the nanocomposites with surface treated SiO₂ (O-SiO₂) nanoparticles substantially increased with increasing O-SiO₂ content from 0 to 3 wt%. Scanning electron microscopy imaging revealed that the nanocomposites with O-SiO₂ nanoparticles exhibited numerous tortuous cracks and ridges, indicating ductile deformation prior to fracturing.

Keywords: Poly(lactic acid), Poly(butylene succinate), Silicon Dioxide, Surface Modification, Impact Strength

INTRODUCTION

Traditional petroleum-based polymers, including polyethylene, poly(vinyl chloride), and acrylonitrile butadiene styrene, have been widely used in several industries. However, these polymers cannot be naturally decomposed, causing serious environmental pollution. Reducing this pollution, necessitates the development of biodegradable and renewable biopolymers to replace the previously listed non-degradable petroleum-based polymers [1-4].

Biopolymers are prepared from renewable resources, and poly(lactic acid) (PLA) is the most popular biodegradable aliphatic polyester. PLA exhibits excellent biodegradability, biocompatibility, and nontoxic properties, and is widely used in packaging, textiles, cutlery, and biomedical applications [5-11]. Although PLA exhibits moderate physical properties compared with other biopolymers, commercial applications are limited by its high production cost and brittle nature.

In general, PLA can be toughened by plasticization, copolymerization, blending, and compounding. In the blending method, PLA is mixed with a biodegradable polymer to improve toughness and maintain biodegradability [12-14]. Among biodegradable polymers, poly(butylene succinate) (PBS) is a biodegradable aliphatic polyester that is inexpensive, biodegradable, processable, thermally stable, and chemically resistant. PBS also exhibits a low glass transition temperature of -45 to -5 °C, imparting it with substantial flexibility [15-20].

In the compounding method, PLA is mixed with nano-sized particles to create new biopolymer composites that exhibit excellent characteristics, including high toughness, good stiffness, and a high

heat distortion temperature. Commonly used nano-sized particles include clay, silicon dioxide (SiO₂) nanoparticles, nanocellulose, and carbon-based nanoparticles [14].

Recently, nanoparticles have been widely used as compatibilizers to enhance the interfacial interaction between polymer blends. Tan et al. [21] prepared PLA/PBS/organically modified clay nanocomposites via volume pulsating elongation flow. Their results showed that the clay mainly dispersed in PBS and located at the interface of PLA/PBS blends. The addition of clay suppressed the aggregation of PBS, resulting in a decrease of the PBS diameter in the PLA/PBS blends. Wu et al. [22] studied the effect of nitrogen-doped graphene (NG) on the morphology and properties of PBS/PLA blends. Their results indicated that the domain size of the dispersed PLA phase in PBS/PLA blends decreased upon the addition of NG, indicating a compatibilization effect of NG on the immiscible blends. Transition electron microscopy (TEM) micrographs showed that the NG mainly dispersed in the PBS matrix, whereas a small amount was located in the PLA phase.

In this study, bio-based PLA/PBS/SiO₂ nanocomposites were fabricated by solution blending. The surface of the SiO₂ nanoparticles was treated with oleic acid. The influence of the SiO₂ amount on the thermal and flexural properties, impact strength, and morphology of the composites was investigated by thermogravimetric analysis (TGA), mechanical testing, Izod impact testing, and field-emission scanning electron microscopy (FE-SEM).

EXPERIMENTAL

1. Materials

PLA (4032 D, weight-average molecular weight of 100,000 g/mol) as the polymer matrix was obtained from NatureWorks Co., USA. PBS (weight-average molecular weight of 50,000) was supplied by Eko New Materials Co., Ltd. (USA). SiO₂ nanoparticles (AEROSIL

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200) with a specific surface area of 200 m²/g and an average particle size of 50 nm were purchased from Degussa (Germany) for use as a reinforcing agent. Oleic acid was supplied by Tianjin Damao Chemical Reagent Factory (China) and used to modify the SiO₂ nanoparticles. Dichloromethane was obtained from Zhangjiagang Free Trade Zone Vibration Good Trade Co., Ltd. (China) and used as a solvent.

2. Surface Modification of SiO₂ Nanoparticles

Oleic acid (2 g) was dissolved in deionized water (100 mL) and SiO₂ (20 g) was subsequently added to the oleic acid solution. Ammonia solution (0.1 mol/L) was added to adjust the pH of the solution in the range of 8-9, and the mixture was stirred continuously at 30 °C for 10 h. The surface-treated SiO₂ nanoparticles were obtained after filtering, washing, and drying. The oleic acid-treated SiO₂ was denoted as O-SiO₂.

3. Sample Preparation

The PLA/PBS/SiO₂ ratio was varied from 80/20/0 to 80/20/4. PLA, PBS, and SiO₂ were dried at 60 °C for 24 h before use. PLA, PBS, and SiO₂ were added to dichloromethane and mixed at 30 °C for 3 h. The dichloromethane was subsequently removed by heat treatment at 180 °C for 4 h. The mixture was then injected into a mold and compression-molded at 180 °C and 4 MPa.

4. Characterization and Measurements

The surface properties of the SiO₂ nanoparticles before and after surface treatment were investigated using X-ray photoelectron spectroscopy (XPS, Thermo scientific, K-Alpha). The thermal properties of the SiO₂ nanoparticles before and after surface treatment were evaluated by TGA (TA Instruments, Q50) from 30 to 800 °C at a heating rate of 10 °C/min under a nitrogen flow of 30 mL/min. The morphology of the SiO₂ nanoparticles before and after surface treatment was examined by FE-SEM (Hitachi, SU 8010) and field-emission-transmission electron microscopy (FE-TEM, JEOL, JEM2100).

The thermal properties of the PLA/PBS/SiO₂ nanocomposites were investigated by TGA (TA Instruments, Q50) from 30 to 500 °C at a heating rate of 10 °C/min under a nitrogen flow of 30 mL/min.

Flexural tests of the nanocomposites were carried out according to methodology specified in standard ASTM D790-86. The impact strength of the composites was measured using an Izod impact tester (TP04G-AS1) according to the GB/T 1843-2008 standard method.

The morphology of the nanocomposites after the impact strength tests was investigated by FE-SEM (Hitachi, SU 8010). The dispersion state of the SiO₂ nanoparticles in PLA/PBS/SiO₂ nanocomposites was examined by FE-TEM (JEOL, JEM2100).

RESULTS AND DISCUSSION

1. Surface Modification of the SiO₂ Nanoparticles

Fig. 1 shows the XPS spectra of the SiO₂ nanoparticles before and after surface treatment. The characteristic peak of O1s, which was observed at 533.8 eV, decreased substantially in intensity after the surface treatment with oleic acid. This decrease in intensity is attributed to the hydroxyl group on the SiO₂ surfaces reacting with oleic acid during the surface treatment [23,24].

Fig. 2 shows the TGA thermograms of the SiO₂ nanoparticles before and after the surface treatment. The weight loss at 800 °C

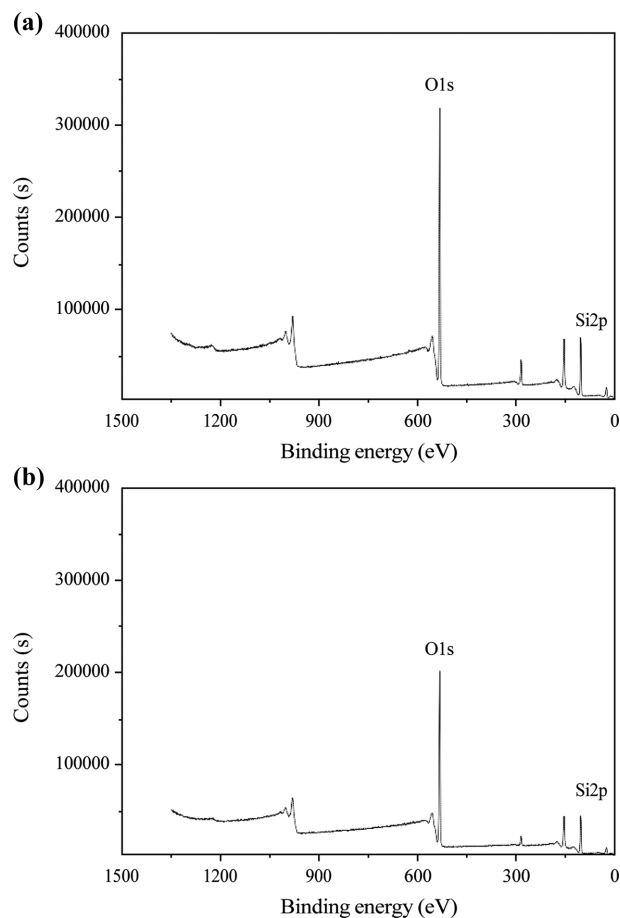


Fig. 1. XPS spectra of SiO₂ nanoparticles (a) before and (b) after surface modification.

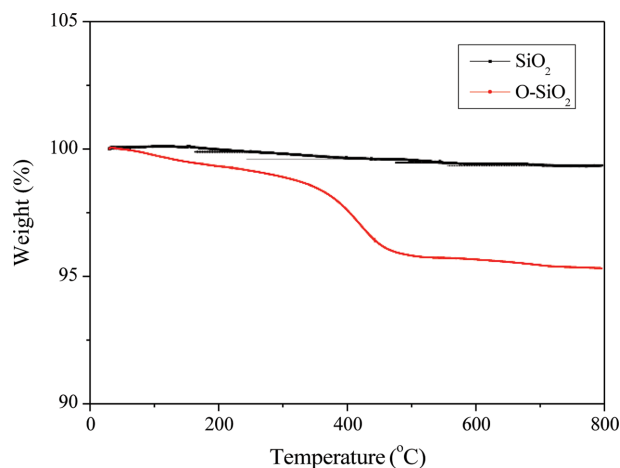
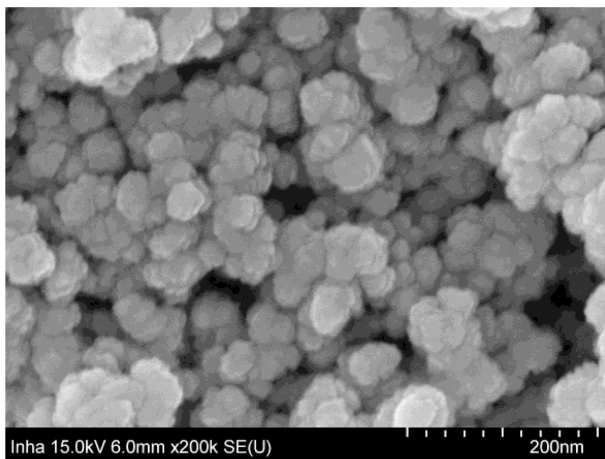


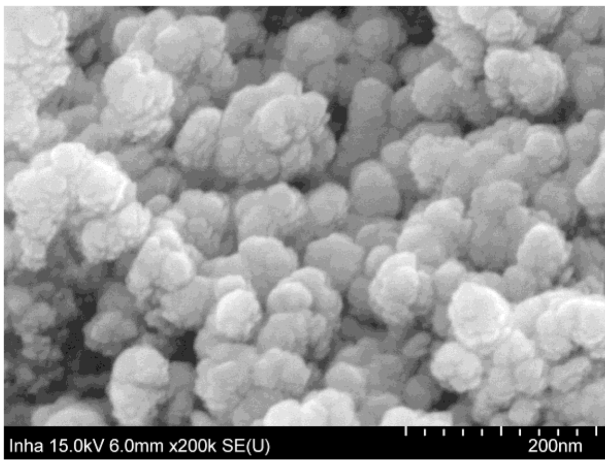
Fig. 2. TGA thermograms of SiO₂ nanoparticles (a) before and (b) after surface modification.

was calculated to be 0.64 and 4.66 wt% for the SiO₂ nanoparticles before and after surface treatment, respectively. These results indicate that the oleic acid content on the surface of O-SiO₂ was approximately 4 wt% [25,26].

The morphology of the SiO₂ nanoparticles before and after surface treatment was observed by SEM and TEM; the correspond-



(a)



(b)

Fig. 3. SEM micrographs of SiO₂ nanoparticles (a) before and (b) after surface modification (magnification of 200,000).

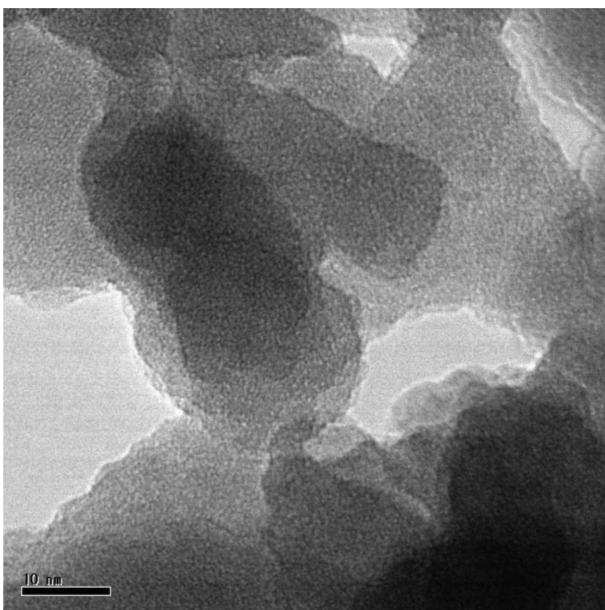


Fig. 4. TEM image of O-SiO₂ nanoparticles.

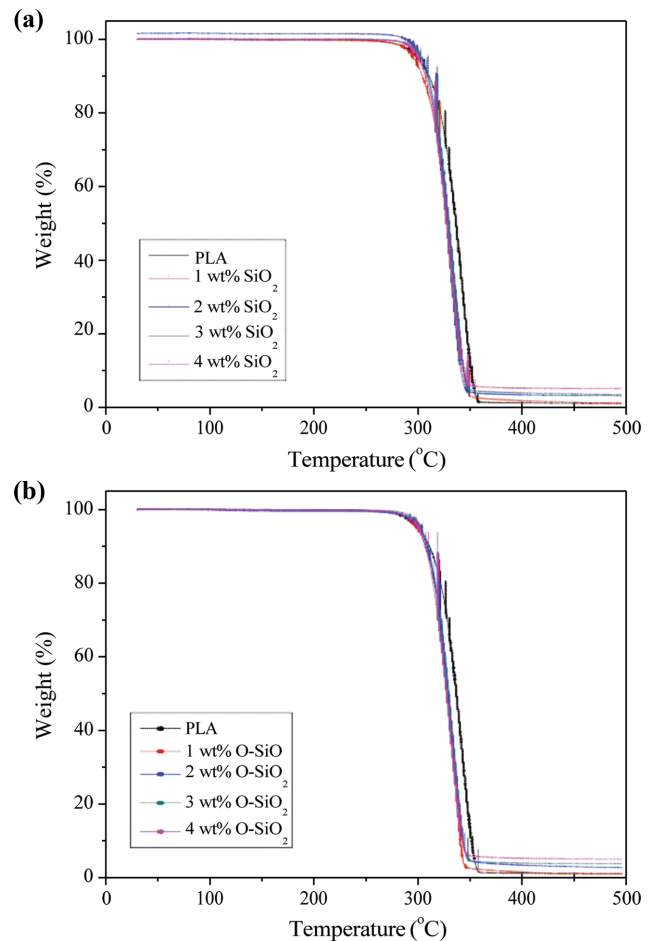


Fig. 5. TGA thermograms of PLA/PBS/SiO₂ nanocomposites as a function of SiO₂ content.

ing images are shown in Figs. 3 and 4, respectively. After surface treatment, the surface of the O-SiO₂ nanoparticles was coated with a thin layer compared with that of the pristine SiO₂ nanoparticles, indicating the attachment of organic moieties on the SiO₂ surfaces [26,27]. Thus, surface treatment with oleic acid introduced organic functional groups onto the surface of the SiO₂ nanoparticles.

2. Thermal Properties of the PLA/PBS/SiO₂ Nanocomposites

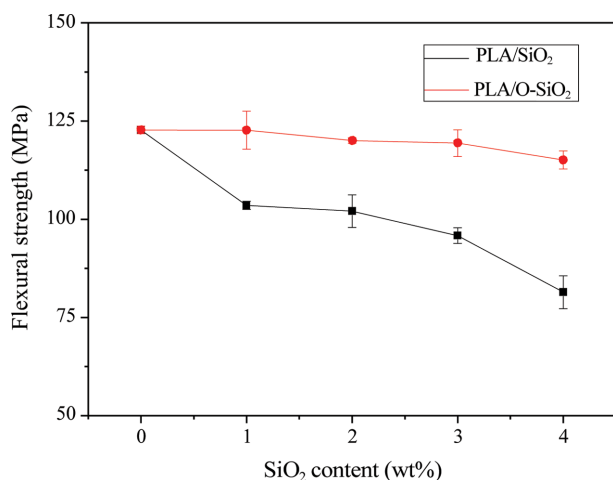
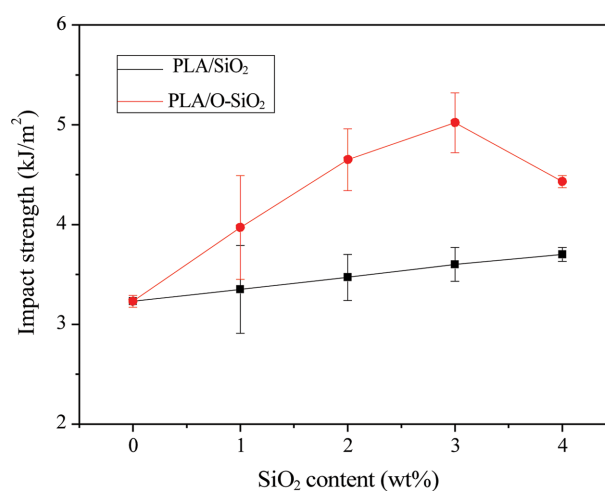
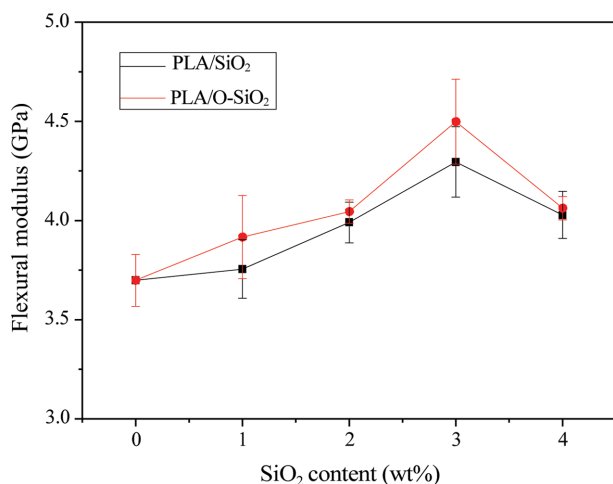
Fig. 5 shows the thermal properties of the PLA/PBS/SiO₂ nanocomposites. The thermal stability factors, including the decomposition temperature ($T_{5\%}$) for weight loss of 5% and char at 500 °C, were calculated from the TGA thermograms [28-30]; the results are summarized in Table 1. The $T_{5\%}$ value of the nanocomposite was similar to that of the neat PLA. The char value of the nanocomposites at 500 °C increased with increasing SiO₂ content. These results indicate that the introduction of the SiO₂ nanoparticles did not affect the thermal stability of the PLA polymer.

3. Flexural Properties of the PLA/PBS/SiO₂ Nanocomposites

The mechanical properties of the PLA/PBS/SiO₂ nanocomposites were studied using flexural strength and elastic modulus measurements. Fig. 6 shows the flexural strength of the PLA/PBS/SiO₂ nanocomposites. The flexural strength of the nanocomposites containing pristine SiO₂ nanoparticles decreased significantly from 122.7 to 81.4 MPa with increase in SiO₂ content from 0 to 4 wt%.

Table 1. Thermal stability factors of PLA/PBS/SiO₂ nanocomposites obtained from TGA thermograms

PBS content (wt%)	SiO ₂ content (wt%)	O-SiO ₂ content (wt%)	T5% (°C)	Char at 500 °C (%)
0	0	0	300.1	1.24
20	1	0	300.8	1.33
20	2	0	300.9	3.39
20	3	0	303.2	3.63
20	4	0	300.5	5.26
20	0	1	298.6	1.17
20	0	2	300.8	2.90
20	0	3	302.8	3.95
20	0	4	300.0	5.18

**Fig. 6. Flexural strength of PLA/PBS/SiO₂ nanocomposites as a function of SiO₂ content.****Fig. 8. Impact strength of PLA/PBS/SiO₂ nanocomposites as a function of SiO₂ content.****Fig. 7. Elastic modulus of PLA/PBS/SiO₂ nanocomposites as a function of SiO₂ content.**

However, the flexural strength of the nanocomposites containing O-SiO₂ nanoparticles only slightly decreased from 122.7 to 115.1 MPa when the O-SiO₂ content increased. This is attributed to the improved interfacial adhesion between the O-SiO₂ nanoparticles and PLA matrix in the PLA/SiO₂ nanocomposites [31,32].

Fig. 7 shows the elastic modulus of the PLA/PBS/SiO₂ nanocomposites. The elastic moduli of the nanocomposites with pristine SiO₂ and O-SiO₂ nanoparticles increased with increasing SiO₂ content from 0 to 3 wt%. The elastic moduli of the PLA/PBS/SiO₂ nanocomposites with pristine SiO₂ and O-SiO₂ nanoparticles were 4.3 and 4.5 GPa, which are 16% and 22% higher than that of pristine PLA, respectively.

4. Impact Strength of the PLA/PBS/SiO₂ Nanocomposites

Fig. 8 shows the impact strength of the PLA/PBS/SiO₂ nanocomposites. The impact strength of the nanocomposites containing pristine SiO₂ nanoparticles increased slightly from 3.2 to 3.7 kJ/m² with increasing SiO₂ content from 0 to 4 wt%. The impact strength of the nanocomposites containing O-SiO₂ nanoparticles significantly increased from 3.2 to 5 kJ/m² when the O-SiO₂ content was increased from 0 to 3 wt%, resulting in a 56% increase over that of the pristine PLA and a 35% increase over that of the nanocomposites with neat SiO₂ nanoparticles. This result is attributed to the O-SiO₂ nanoparticles functioning as a compatibilizer and improving the interfacial interaction between the PLA and PBS polymers [21,22]. The increase in impact strength was also due to numerous cracks created in the PLA matrix by the SiO₂ nanoparticles, which enabled the absorption of more external energy [33,34].

The impact strength of the nanocomposites containing O-SiO₂

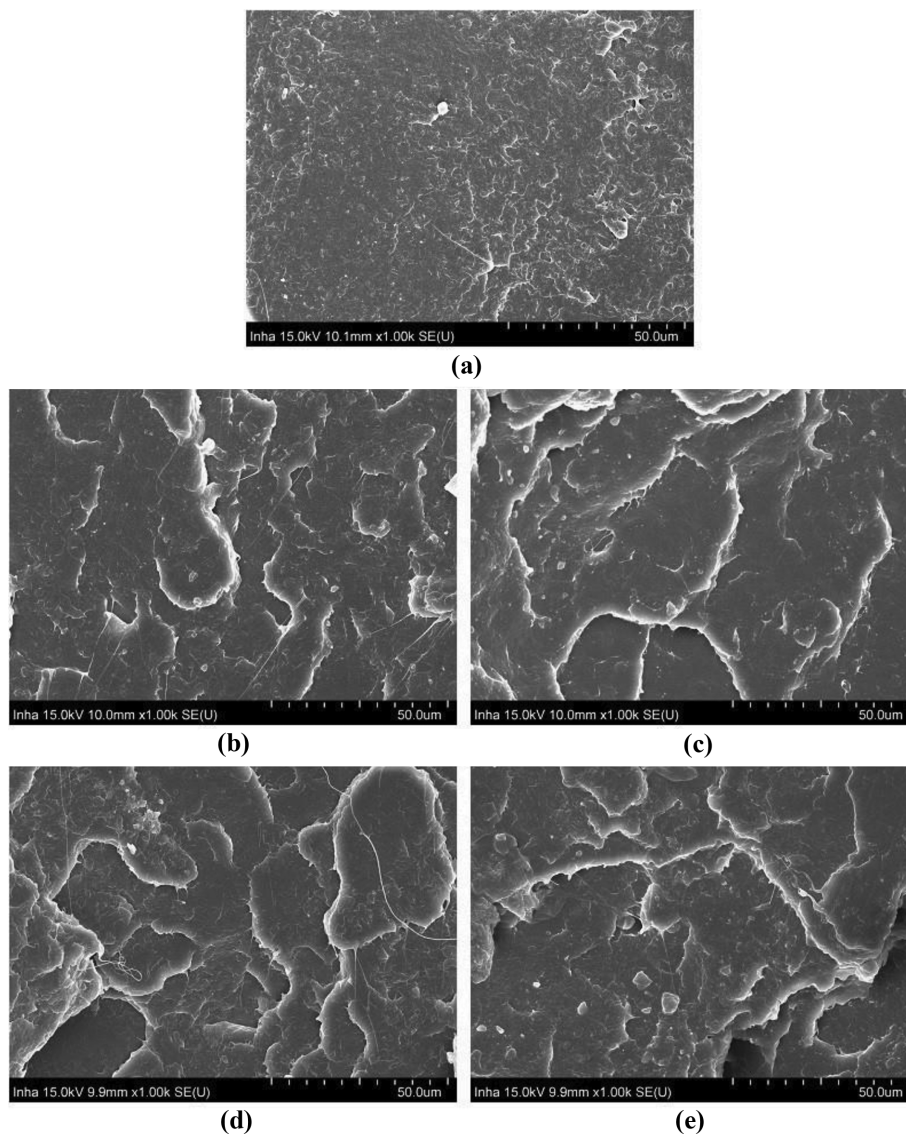


Fig. 9. SEM micrographs of PLA/PBS/SiO₂ nanocomposites after impact strength tests: (a) Pristine PLA; (b) 1 wt% SiO₂; (c) 3 wt% SiO₂; (d) 1 wt% O-SiO₂; (e) 3 wt% O-SiO₂ (magnification of 1,000).

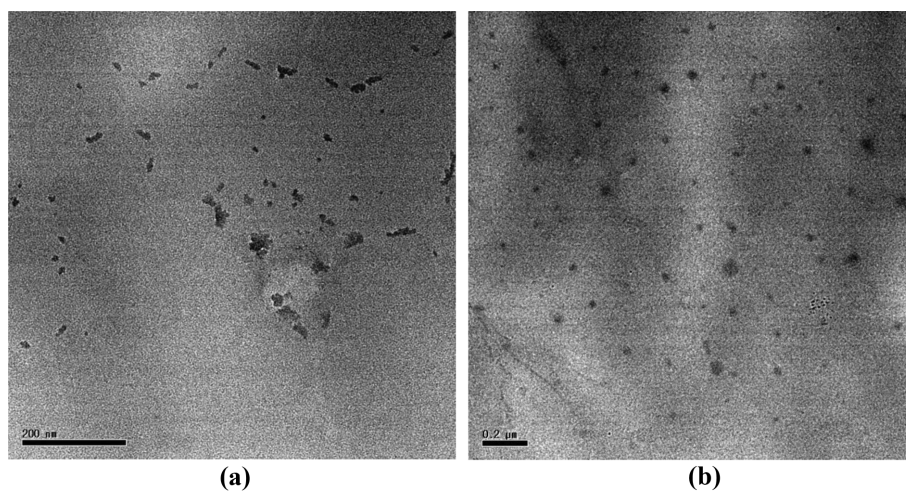


Fig. 10. TEM images of PLA/PBS/SiO₂ nanocomposites: (a) Aggregated SiO₂ nanoparticles; (b) well distributed O-SiO₂ nanoparticles.

nanoparticles slightly decreased when the O-SiO₂ content increased from 3 to 4 wt%. This result is attributed to the O-SiO₂ nanoparticles aggregating at high O-SiO₂ contents.

5. Morphology of the PLA/PBS/SiO₂ Nanocomposites

The toughness behavior of the PLA/PBS/SiO₂ nanocomposites was further characterized by SEM. Fig. 9 shows SEM images of the PLA/PBS/SiO₂ nanocomposites after the impact strength tests. As shown in Fig. 9(a), the neat PLA exhibits a flattened morphology, indicating a brittle fracture surface [35]. By contrast, the SEM images of the nanocomposites with pristine SiO₂ nanoparticles show a rough morphology, indicating ductile deformation prior to fracturing (Figs. 9(b) and (c)). The SEM images of the nanocomposites with O-SiO₂ nanoparticles show numerous tortuous cracks and ridges, indicating ductile deformation prior to fracturing (Figs. 9(d) and (e)) [34,36], accounting for the high impact strength of the PLA/PBS/SiO₂ nanocomposites.

The dispersion state of the SiO₂ nanoparticles in the PLA/PBS/SiO₂ nanocomposites was examined by TEM; the corresponding micrographs of the nanocomposites are shown in Fig. 10. As shown in Fig. 10(a), the SiO₂ nanoparticles are aggregated in the nanocomposites. After the surface modification, SiO₂ nanoparticles were well distributed in the nanocomposites, as shown in Fig. 10(b).

CONCLUSIONS

The thermal and flexural properties, impact strength, and morphology of the PLA/PBS/SiO₂ nanocomposites were examined. The thermal stability of the nanocomposite was similar to that of the neat PLA. The flexural strength of the nanocomposites slightly decreased after the surface treatment of the SiO₂ nanoparticles. The nanocomposites showed a maximum value of elastic modulus at 3 wt% SiO₂. The impact strength of the nanocomposites containing O-SiO₂ nanoparticles substantially increased with increasing O-SiO₂ content from 0 to 3 wt%, reaching a value 56% greater than that of the neat PLA and 35% greater than that of the nanocomposites containing pristine SiO₂ nanoparticles. The results indicate that improved impact strength of PLA was achieved without sacrificing the thermal and mechanical properties of the nanocomposites.

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